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ION BEAM CHANNELING AND HYPERFINE INTERACTION ANALYSIS FOR THE CHARACTERIZATION OF STOICHIOMETRY AND ANTI-SITE POPULATION IN LiNbO_3

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ABSTRACT

Combined high precision RBS/channeling, NRA/channeling and perturbed angular studies were used to characterize near-stoichiometric and congruent LiNbO_3 single crystals. In congruent LiNbO_3 two non-equivalent Li sites are distinguished whereas in near-stoichiometric LiNbO_3 only one of them exists. A comparison of the experimental results with Monte Carlo simulations was performed to check the validity of the different models for the structure of congruent lithium niobate crystals.

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Ion Beam Channeling and Hyperfine Interaction Analysis for the characterization of stoichiometry and anti-site population in LiNbO₃

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1. Introduction

Lithium niobate (LiNbO₃) is an important material for applications in optoelectronic and integrated optics devices. Many of its properties are influenced by intrinsic defects due to deviations from the stoichiometry and extrinsic defects introduced by dopants. Crystals grown by the Czochralski technique from a congruently melting composition have a Li to Nb ratio [Li]/[Nb]=0.945, which differs from the 1:1 ratio inferred from the chemical formula LiNbO₃. The Li deficiency leads to a defect structure for which several models have been proposed. According to the most common model by Abrahams and Marsh [1], based on X-ray diffraction measurements, Nb occupies about 5.9% of the Li sites forming niobium antisites (Nb_{Li}). In order to obtain charge compensation, 4.8% of the Nb sites have to be vacant. In contrast, Lerner et al. [2] suggested that the Nb lattice may not contain any vacancies and the number of Nb antisites is limited to the excess Nb fraction of 1.2%. Charge compensation would then be achieved by a sufficient number of Li vacancies. An alternative model, first proposed by Nassau and Lines [3], explains the defect structure of congruent LiNbO₃ by the existence of stacking fault type defects. Recent computational results obtained by Donnerberg et al. [4] indicate that these defects could be ilmenite-type stacking sequences probably forming substantial intergrowths.

The recent characterization of bulk stoichiometric LiNbO_3 crystals grown from melts containing 6% K_2O [5] showed a significant change of some properties, such as decrease in the linewidths of ^{93}Nb NMR, in the Fe^{3+} electron paramagnetic resonance and in the Raman line of the E phonon at 153 cm^{-1} [5], but no correlation to the defect structure of congruent LiNbO_3 was made.

Ion beam channeling and hyperfine interaction measurements are extremely sensitive to structural properties of single crystals and yield complementary information [6]. Using these techniques, a comparative study of the defect properties of both near-stoichiometric and congruent lithium niobate crystals is presented in this work.

2. Experimental details

Near-stoichiometric LiNbO_3 crystals were grown at the Universidad Autonoma of Madrid using the Czochralski method from a melt containing grade I Johnson-Mathey powder to which 4.6% K_2O were added. Congruent LiNbO_3 crystals, also grown by the Czochralski method, were obtained from Munich and Budapest. Samples were cut both parallel and perpendicular to the c-axis and polished. PIXE analysis of the near-stoichiometric plates indicates that potassium is not incorporated in the crystals down to the detection level of 100 ppm. The optical absorption measurements show an optical absorption edge ($\alpha=20\text{cm}^{-1}$) at 314 nm for the near-stoichiometric crystal, and at 320 nm for a congruent one. From the calibration curve given in Ref. 7 a Li to Nb ratio of $[\text{Li}]/[\text{Nb}]=0.979$ for the near stoichiometric crystal was derived.

The RBS/C and NRA/C experiments were carried out at the 3.1 MeV Van de Graaff accelerator at Sacavém. The orientation of the crystals was checked using a 1.6 MeV He^+ beam. The channeling investigations of the Nb-RBS and the Li-NRA with the $^7\text{Li}(p,\alpha)^4\text{He}$ reaction were performed using both 1.5 and 1.6 MeV H^+ beams. The α particles with an energy of about 7.5 MeV released in this reaction can be easily distinguished from the backscattered protons. The charged particles were detected at an angle near 180° using an annular silicon surface barrier detector with a resolution of 18 keV.

For the hyperfine interaction measurements, radioactive $^{111\text{m}}\text{Cd}$ was implanted at 60 keV energy using the ISOLDE/CERN isotope separator, to doses of 5×10^{12} at/cm². The projected range of the Cd⁺ ions is 23 nm and the peak concentration is 1×10^{18} Cd/cm³, i.e., about 0.02 at%. After implantation the samples were annealed in air at 873 K for 30 min. and slowly cooled down to the measuring temperature of 295 K. The e⁻- γ PAC experiments were performed using the well-known 151-245 keV cascade from the decay of $^{111\text{m}}\text{Cd}$ with an anisotropy $b_2A_{22}=+0.230$. The 245 keV state has spin $I=5/2$, half-life of $T_{1/2}=84$ ns and quadrupole moment of $|Q|=0.81(13)\text{b}$ [8]. The experimental set-up consists of two β spectrometers of the Siegbahn type for the detection of conversion electrons and two BaF₂ scintillators for γ detection, arranged in a plane at 90° intervals [9], with a time resolution of 1.0 ns (FWHM). From the coincidence spectra $N(\theta, t)$, where θ is the angle between detectors and t is the time delay between events, the time differential anisotropy $R(t)=2.[N(180^\circ, t)-N(90^\circ, t)]/[N(180^\circ, t)+2.N(90^\circ, t)] \sim b_2A_{22}G_{22}(t)$ was calculated. $G_{22}(t)$ is the perturbation function which describes the modulation of the angular correlation. For a cascade with $I=5/2$ three frequencies $\omega_n=C_n(\eta)v_Q$ are observed, giving

$$G_{22}(t)=\sum_{n=0}^3 S_{2n} \cos (C_n(\eta)v_{Qt}) \exp (-\frac{1}{2} \delta C_n(\eta)v_{Qt})$$

The quadrupole coupling constant $v_Q=eQV_{ZZ}/h$ and the asymmetry parameter $\eta = (V_{xx}-V_{yy})/V_{zz}$, which can be deduced from the frequency factors $C_n(\eta)$, contain information about the magnitude of the principal component V_{zz} and the asymmetry of the EFG. The exponential factor allows for a Lorentzian distribution of EFGs around a mean value. Such a distribution can be caused e.g. by different lattice defects in the vicinity of the probe atoms. The S_{2n} coefficients, which give the amplitudes for each frequency, can be calculated for polycrystalline as well as for single crystalline samples. If fractions f_i of the total number of radioactive probes are in different but unique sites experiencing different EFGs the observed spectra is described by a sum over several perturbation functions, $f_i.G_{22}^i(t)$, with the sum of the f_i normalized to unity.

3. Results and Discussion

3.1 RBS/C and NRA/C

Figs.1a) and b) show NRA spectra of the ${}^7\text{Li}(p,\alpha){}^4\text{He}$ reaction for 1.5 MeV protons aligned with the $\langle 11\bar{2}0 \rangle$ axis and in random direction, for congruent and stoichiometric LiNbO_3 , respectively. The minimum yield derived from the aligned spectrum, near the surface (1.5 μm), is remarkably lower for the stoichiometric crystal, $\chi_{\text{min}}=53\%$, than for the congruent samples: $\chi_{\text{min}}=74\%$ (Budapest) and $\chi_{\text{min}}=73\%$ (Munich, not shown here). From recent experiments on the dependence of the Li-NRA minimum yield on the Mg concentration [10] it is known that even for high Mg concentrations no χ_{min} lower than 60% is observed for this axis. Both observations show that the Li sublattice in the case of near-stoichiometric LiNbO_3 must be remarkably closer to the perfect arrangement.

Additionally, the aligned spectrum for the stoichiometric sample exhibits a much higher yield than the random spectrum for α -particles emerging from higher depths. This can be attributed to a significantly lower stopping power in the axial direction for the stoichiometric crystal, which is expected for channeling in a perfect lattice. The resulting higher proton energy at a given depth and the corresponding higher nuclear cross section for the ${}^7\text{Li}(p,\alpha){}^4\text{He}$ reaction leads to an enhanced yield.

Figs.2a) and b) show Li and Nb angular scans for the (0001) plane and the $\langle 11\bar{2}0 \rangle$ axis along the (0001) plane, respectively. For planar channeling, the Li-NRA peaks observed for the stoichiometric and congruent crystals differ significantly in width and height. This hints to substantial differences in the structure of the Li sublattices. Also the Li-NRA angular scan for the $\langle 11\bar{2}0 \rangle$ axis along the (0001) plane corroborates the differences between both crystals.

Computer simulations using the CASSIS code [11] show that in the case of stoichiometric lithium niobate the (0001) angular scan can be described assuming a perfect crystal lattice, as shown in Fig. 3. For the congruent crystals, simulation results obtained assuming different defect models previously discussed [1-4] are depicted in Fig.4. The experimental data are also included for comparison.

The models, that correlate the defect structure only to the niobium antisites, just yield a decrease of the Li-NRA peak height (Fig. 4a and 4b), but do not reproduce the experimentally observed broadening. Only the assumption of two different lithium sites as derived from the occurrence of cation stacking faults with inversion symmetry (ilmenite type) results in both a decrease and broadening of the Li-NRA peak (Fig. 4c). In this case assuming a fraction of 2% dilute randomly distributed stacking faults yields a better fit to the experimental data is obtained.

The angular scans measured for Nb-RBS almost overlap for both congruent and stoichiometric samples. For all defect models only a minor influence on the Nb angular scans is expected, since a small fraction of Nb in Li sites (either 1.2% or 5.9%) would only yield a small increase in the Nb-RBS minimum yield. In the case of the stacking fault model the additional Nb site in the c-plane is projected close to the regular one and therefore cannot be easily distinguished by RBS.

3.2 Perturbed Angular Correlation

Figs. 5a) and b) show the PAC spectra and respective Fourier analysis obtained for the congruent and the stoichiometric sample, respectively. The c-axis of the single crystals was in the detectors plane at 45° with two detectors.

The amplitude of the time differential anisotropy $R(t)$ at $t \sim 200$ ns is clearly larger for the stoichiometric crystal than for the congruent one. As discussed previously [12] for measurements in a congruent crystal, ^{111}Cd replaces Li but the PAC spectra cannot be described by only one quadrupole interaction frequency. In fact, the Fourier analysis of the spectra shows a second frequency with a value only slightly higher than the main component. The best fit to the data was obtained considering two nearby Li-site frequencies, $\nu_Q^I = 191(2)$ MHz and $\nu_Q^{II} = 205(2)$ MHz, with relative site occupancies of 71(5)% and 29(5)%, respectively. Due to the symmetry properties of the LiNbO_3 lattice the EFG expected at the Li site should have axial symmetry (i.e., $\eta = 0$). However, the spectra can only be described assuming $\eta > 0$ for both sites, $\eta^I = 0.11(2)$ and $\eta^{II} = 0.16(4)$. In contrast, the Fourier analysis of the spectra taken for the stoichiometric crystal shows sharper lines and the data are well described by one frequency only, $\nu_Q = 191(2)$ MHz, and an axially

symmetric EFG, i.e., $\eta=0$. This indicates a higher lattice perfection for the stoichiometric sample. A study of the stoichiometric LiNbO_3 using the ^{93}Nb NMR probe [5] also yielded a sharper resonance line and an absence of angular dependence of its width, which is a characteristic of an axially symmetric EFG. This is in contrast with the congruent case, where the resonance is broad and a strong angular dependence of the width is observed.

4. Conclusions

Ion beam analysis and hyperfine interaction techniques were used to characterize for the first time near-stoichiometric LiNbO_3 crystals. Both techniques show that the lattice of the stoichiometric sample is undisturbed. A comparative study on congruent crystals reveals that in this case two distinct Li sites exist. Computer simulations of the Li-NRA and Nb-RBS angular scans for planar channeling rule out the models assuming a substantial fraction of antisite niobium to be responsible for the structural disorder. A cation stacking fault of the ilmenite type occurring randomly with a fraction of 2% reproduces the lithium sites with the observed relative site occupancies.

Acknowledgments

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References

- [1] S.C. Abrahams and P. Marsh, *Acta Crystallogr. B* 42 (1986) 61
- [2] P. Lerner, C. Legras and J.P. Dumas, *J. Cryst. Growth* 3/4 (1968) 231
- [3] K. Nassau and M.E. Lines, *J. Appl. Phys.* 41 (1970) 533
- [4] H. Donnerberg, S.M. Tomlinson, C.R.A. Catlow and O.F. Schirmer, *Phys. Rev. B* 40 (1989) 11909
- [5] G.I. Malovichko, V.G. Grachev, E.P. Kokanyan, O.F. Schirmer, K. Betzler, B. Gather, F. Jermann, S. Klauer, U. Schlarb and M. Wöhlecke, *Appl. Phys. A* 56 (1993) 103
- [6] J.C. Soares, *Nucl. Instr. and Meth. B* 64 (1992) 215
- [7] I. Földvari, K. Polgár, R. Voszka, R.N. Balasanyang, *Crystal. Res. Tech.* 19 (1984) 1659
- [8] P. Herzog, K. Freitag, M. Reuschenbach and H. Walitzi, *Z. Phys.* A294 (1980) 13
- [9] J.G. Marques, J.G. Correia, A.A. Melo, M.F. da Silva, J.C. Soares and ISOLDE Collaboration, *Nucl. Instr. and Meth. B* 99 (1995) 645
- [10] A. Kling, D. Kollwe and B.C. Grabmaier, *Nucl. Instr. and Meth. B* 85 (1994) 490
- [11] A. Kling, *Nucl. Instr. Meth. B*, in press.
- [12] B. Hauer, R. Vianden, J.G. Marques, N.P. Barradas, J.G. Correia, A.A. Melo, J.C. Soares, E. Diéguez and F. Agulló-López, *Phys. Rev. B* 51 (1995) 6208

Figure Captions

Fig. 1: NRA spectra (α particles produced in the ${}^7\text{Li}(p,\alpha){}^4\text{He}$ reaction) for incident 1.5 MeV H^+ beam in a random direction and aligned with the $\langle 11\bar{2}0 \rangle$ axis for (a) congruent LiNbO_3 , (b) stoichiometric LiNbO_3 .

Fig. 2: NRA and RBS channeling dips for (a) the (0001) plane and (b) $\langle 11\bar{2}0 \rangle$ axis along the (0001) plane. The solid lines are drawn to guide the eyes.

Fig. 3: Comparison of experimental results and computer simulation for the Li-NRA (0001) angular scan in the case of stoichiometric lithium niobate.

Fig. 4: Comparison of experimental results and computer simulation for the Li-NRA (0001) angular scan assuming different defect models for congruent lithium niobate: a) 5.9% niobium antisites, b) 1.2% niobium antisites and c) 2% ilmenite type stacking faults.

Fig. 5: Time dependent anisotropy of the 151-245 keV $e^- \gamma$ cascade of ${}^{111}\text{Cd}$ for (a) congruent LiNbO_3 , (b) stoichiometric LiNbO_3 . The c-axis of the single crystals was in the detectors' plane at 45° with two detectors.